

# Methane and Carbon Dioxide in Dual-Porosity Organic Matter: Molecular Simulations of Adsorption and Diffusion

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Shale gas, which predominantly consists of methane, is an important unconventional energy resource that has had a potential game-changing effect on natural gas supplies worldwide in recent years. Shale is comprised of two distinct components: organic material and clay minerals, the former providing storage for hydrocarbons and the latter minimizing hydrocarbon transport. The injection of carbon dioxide in the exchange of methane within shale formations improves the shale gas recovery, and simultaneously sequesters carbon dioxide to reduce greenhouse gas emissions. Understanding the properties of fluids such as methane and methane/carbon dioxide mixtures in narrow pores found within shale formations is critical for identifying ways to deploy shale gas technology with reduced environmental impact. In this work, we apply molecular-level simulations to explore adsorption and diffusion behavior of methane, as a proxy of shale gas, and methane/carbon dioxide mixtures in realistic models of organic materials. We first use molecular dynamics simulations to generate the porous structures of mature and overmature type-II organic matter with both micro- and mesoporosity, and systematically characterize the resulting dual-porosity organic-matter structures. We then employ the grand canonical Monte Carlo technique to study the adsorption of methane and the competing adsorption of methane/carbon dioxide mixtures in the organic-matter porous structures. We complement the adsorption studies by simulating the diffusion of adsorbed methane, and adsorbed methane/carbon dioxide mixtures in the organic-matter structures using molecular dynamics.

**Keywords:** CO<sub>2</sub> sequestration; grand canonical Monte Carlo; molecular dynamics; shale formation; shale gas recovery

# 1 Introduction

Shale gas, which predominantly consists of methane, is an unconventional energy source that is expected to play an imminent role in the energy market of the immediate future [1]. Shale rocks are composed of organic and inorganic matter with pore sizes ranging from the nano- to the mesoscale. The organic matter (OM) provides storage for hydrocarbons, while the inorganic matter (clay minerals) minimizes hydrocarbon transport. The OM is primarily comprised of kerogen, but it also contains other compounds such as resins, asphaltenes and heavy hydrocarbons. In shale formations, shale gas is stored as free gas in fractures and macropores, as adsorbed gas on pore surfaces and nanopores, and as dissolved gas in the OM. The sorption amount of the gas depends largely on the total organic carbon (TOC) content [2].

Carbon dioxide exhibits a stronger interaction with shale pore surfaces than methane, where this preferential adsorption of  $\text{CO}_2$  over  $\text{CH}_4$  plays the key role in displacing adsorbed shale gas. On the other hand, due to the stronger  $\text{CO}_2$  interaction, the diffusion of carbon dioxide in shale pores is lower in comparison with methane, which may limit the displacement of the adsorbed shale gas [3]. As such, enhanced shale gas recovery with  $\text{CO}_2$  injection is a promising technique as it combines the advantages of improving the efficiency of shale gas recovery, and reducing greenhouse gas emissions by geological sequestration of carbon dioxide [4].

Understanding the properties of fluids such as methane and methane/carbon dioxide mixtures in the narrow shale pores is critical for identifying ways to deploy shale gas technology with reduced environmental impact. Laboratory experiments such as the adsorption-desorption method [5], the differential scanning calorimetry method [6], the diffusion method [7], and the nanochannel chips method [8] can provide only limited information about the adsorption and diffusion behavior of fluids in shale pores because typically the experiments are performed under low to medium temperatures and pressures, which differ from realistic reservoir conditions [9]. Molecular-level simulations are important alternatives and complementary tools to experimental measurements since they can provide detailed information about the microscopic mechanism of the gas adsorption/desorption and diffusion processes in the shale matrix at realistic reservoir conditions [10]. Simulation methods for modeling gas adsorption and diffusion in porous materials are well established, which include grand canonical Monte Carlo (GCMC), molecular dynamics (MD) [11, 12], and hybrid GCMC-MD techniques such as dual control volume grand canonical MD [13].

The level of realism of molecular simulations of shale gas adsorption and diffusion critically

depends on building representative models of the shale matrix. Kerogen, which is a main component of the OM, represents a complex material composed of an amorphous porous carbon skeleton, and exhibits significant pore-shape and pore-connectivity variations. The van Krevelen diagram (Fig. 1a), which is a plot of hydrogen-to-carbon (H/C) and oxygen-to-carbon (O/C) atomic ratios, provides a tool to distinguish different types of kerogen in terms of maturity and depositional origin (in parenthesis): type I (lacustrine), type II (marine), type III (terrestrial), and type IV (originating from residues) [14, 15].

Kerogen and its pores can be modelled by (i) constitutive, (ii) mimetic, (iii) reconstruction, and (iv) representative molecular models [16]. The constitutive models include graphitic slit pores and cylindrical pores, such as carbon nanotubes [17, 18, 19]. The mimetic models are built using reactive force fields and quenched MD [20, 21]. The reconstruction models also employ reactive force fields along with a MD-hybrid reverse Monte Carlo method to generate kerogen models by matching experimental and simulated structural properties, such as C-C pair distribution functions [22, 23]. The representative molecular models involve the construction of a model molecule of kerogen based on available experimental data (elemental analysis and scattering data), and prior knowledge of the molecular fragments that likely exist in kerogen. The latter models were pioneered by Ungerer et al. [24] who published a number of kerogen molecules/fragments of smaller sizes, which represented different kerogen types at different maturation stages, and constructed them using experimental data [25] and prior knowledge about the molecular fragments. Most kerogen models neglect the effect of adsorption-induced deformations by considering kerogen as a rigid, non-deformable matrix [26]. However, flexible kerogen models have been considered, where they exhibited an increase in gas diffusivity with respect to rigid kerogens due to fluctuations in the pore connectivity [27, 28, 29, 30]. Due to their robust properties, the representative molecular models have become a preferential approach for modeling shale matrices with a broad diversity of physical properties such as TOC, density, porosity and permeability, as demonstrated by the growing number of molecular simulation studies found in the literature. For example, Collell et al. [31] performed MD simulations of hydrocarbons permeating through oil-prone type II kerogen, and found the permeation mechanism is purely diffusive. Michalec and Lísal [3], and Vasileiadis et al. [32] simulated adsorption and diffusion of various proxies of shale gas along with CO<sub>2</sub> in overmature type II kerogens with control microporosity. The microporosity was introduced into the kerogen structures by dummy particles of various size to mimic neglected compounds in real OM. In another study, Ho et al. [33] showed that methane release in nanoporous kerogen matrices is characterized by a fast release of pressurised free gas (accounting for less than 50 %

recovery), followed by a slow release of adsorbed gas as the gas pressure decreases. There are still other related molecular-level simulation studies on enhanced shale gas recovery by carbon dioxide and CO<sub>2</sub> sequestration in kerogen matrices of different maturity and porosity, see e.g. [34, 35, 36, 37, 38, 39].

Most molecular models treat the shale OM as a dense porous kerogen matrix [9, 10]. Although kerogen is the abundant species in the OM, the OM also contains other compounds such as asphaltenes, resins, and dissolved species (typically hydrocarbons, water and carbon dioxide), where their absence may be mimicked by introducing dummy particles into the kerogen matrix [3, 26]. An exception is the work by Collell et al. [40] who generated the molecular model of type II OM corresponding to midway through the oil formation process. Based on experimental composition data [2], they generated the OM structure by gradually cooling and compressing the initial low-density random configurations, which contained the representative molecular model of type II kerogen, asphaltene/resin, hydrocarbons, water, and carbon dioxide. The kerogen taken together with the other species provides a more faithful representation of OM with an intrinsic microporous network, as compared to models with solely kerogen.

In this work, we used Collell et al.’s approach [40], and generated two realistic models of a type II OM microporous structure by MD simulation: (i) a mature OM corresponding to the oil formation process; and (ii) an overmature OM corresponding to the gas formation process; see the van Krevelen diagram in Fig. 1a. We then introduced mesoporosity into the OM models by separating the OM microporous structures by a distance of 20 or 30 Å, creating a slit-shaped mesopore with a width corresponding to these distances. This was followed by GCMC simulations of the adsorption behavior of pure methane and equimolar methane/carbon dioxide mixtures in the dual-porosity OM structures. We evaluated the adsorption amount separately in the microporous matrix and the mesopore slit to assess adsorption capacity due to the microporosity and mesoporosity [41]. Equilibrium concentrations from GCMC simulations were then employed in MD simulations to evaluate self-diffusivity of adsorbed methane and carbon dioxide along the adsorption isotherms to provide insight into the interplay between gas adsorption and gas diffusion, and into the molecular mechanism for the potential replacement of adsorbed shale gas by carbon dioxide in the shale OM. The remainder of the paper is organized as follows. Section 2 outlines the generation of the dual-porosity OM structures along with the molecular models and simulation details. In Section 3, we then present and discuss our results on the structure characterization of the OM models, and on the adsorption and molecular diffusion of methane and methane/carbon dioxide

mixtures in the OM structures. Finally, we give our conclusions in Section 4.

## 2 Models and Methodology

### 2.1 Dual-Porosity Organic Matter

Most of the previous molecular-level simulation studies on the adsorption and diffusion of shale gas approximated the OM by dense porous kerogen matrices, while neglecting the presence of other compounds. However, such an approximation holds only for immature OM, which is almost exclusively comprised of immature kerogen. At the early stages of the maturation process of sedimentary OM, carbon dioxide, water, and asphaltene/resin molecules are produced. Then during the later stages of maturation, alkyl chains and cyclics are generated along with a minor fraction of alkanes and aromatics, leading first to crude oils, and then to gases [2, 42].

Here, we consider mature and overmature type II OM, whereby following Collell et al. [40], we assume that the OM are comprised of the kerogen fraction, asphaltene/resin fraction, hydrocarbon fraction, and carbon dioxide/water fraction. Compositions of the OM are based on the experimental data of Tissot and Welte [2], and Kelemen et al. [25], where it is assumed that the atomic balance is preserved during the maturation process. The latter is equivalent to considering that the compounds generated during the diagenesis and catagenesis processes have not been expelled, and that CO<sub>2</sub> has been preserved from reacting with the inorganic matrix. Finally, the distribution of the hydrocarbon fraction corresponds to a typical condensate gas [43].

In the van Krevelen diagram (Fig. 1a), the mature and overmature kerogens are indicated by stars. The mature kerogen fragment has a chemical formula C<sub>242</sub>H<sub>219</sub>O<sub>13</sub>S<sub>2</sub>N<sub>5</sub>, a molecular weight  $M_w = 3468.4$  g/mol, and has a significantly folded structure. The overmature kerogen fragment has a chemical formula C<sub>175</sub>H<sub>102</sub>O<sub>9</sub>S<sub>2</sub>N<sub>4</sub>,  $M_w = 2468.9$  g/mol, and exhibits a flat and rigid structure. The molecular models of both kerogen fragments are shown in Figs. 1b and 1c. The asphaltene/resin fraction corresponds to the hydrocarbon compounds, which are made up of polyaromatic units of lower molecular weight compared to the kerogen. The asphaltene/resin fraction was represented by a generic molecular model with a chemical formula C<sub>26</sub>H<sub>32</sub>O and  $M_w = 360.5$  g/mol [44]. The hydrocarbon fraction was represented by eight compounds lumped together: six paraffins (C<sub>1</sub> to C<sub>4</sub>, C<sub>8</sub> and C<sub>14</sub>) and two representative aromatic compounds (toluene and dimethylnaphtalene) [40, 43]. In addition, both the mature and overmature OM models contain carbon dioxide and water. The compositions of the OM

structures considered in this study are provided in Table S1 of the Supplementary Material along with simulation details to generate the model OM structures.

## 2.2 Molecular Models

We modeled the kerogen fragments, asphaltene/resin molecules, hydrocarbons, water and carbon dioxide using a non-polarizable all-atom force field with the functional form [45]

$$\begin{aligned}
 U &= U_{\text{bonded}} + U_{\text{non-bonded}} \\
 &= \sum_{\text{bonds}} \frac{k_b}{2} (b - b_0)^2 + \sum_{\text{angles}} \frac{k_\theta}{2} (\theta - \theta_0)^2 + \sum_{\text{dihedrals}} k_\phi [1 + \cos(n_\phi \phi)] \\
 &\quad + \sum_{\text{impropers}} k_\psi [1 + \cos(n_\psi \psi)] + \sum_i \sum_{j>i} \left\{ 4\varepsilon_{ij} \left[ \left( \frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left( \frac{\sigma_{ij}}{r_{ij}} \right)^6 \right] + \frac{q_i q_j e^2}{4\pi \epsilon_0 r_{ij}} \right\}
 \end{aligned} \tag{1}$$

where  $b$ ,  $\theta$ ,  $\phi$  and  $\psi$  denote the bond length, bond angle, dihedral angle and improper angle, respectively;  $b_0$  and  $\theta_0$  are the equilibrium bond length and bond angle, respectively;  $k_b$ ,  $k_\theta$ ,  $k_\phi$  and  $k_\psi$  are the force constants;  $n_\phi$  and  $n_\psi$  are the dihedral and improper angle parameters, respectively;  $\varepsilon_{ij}$  and  $\sigma_{ij}$  correspond to the Lennard-Jones (LJ) energy and size parameters, respectively;  $q_k$  is the partial charge on atom  $k$ ,  $e$  is the elementary charge,  $\epsilon_0$  is the permittivity in vacuum, and  $r_{ij}$  is the distance between atoms  $i$  and  $j$ .

For the kerogen fragments and asphaltene/resin molecules, we used the Consistent Valence Force Field (CVFF) [46], which is a standard and reliable force field for simulation of organic molecules. For the hydrocarbons, water, and carbon dioxide, we employed the Optimized Potential for Liquid Simulations (OPLS) [47], the flexible Extended Simple Point Charge (SPC/E) [48, 49], and the flexible Elementary Physical Model (EPM2) [50] force fields, respectively. The OPLS, SPC/E, and EPM2 force fields have been shown to satisfactorily reproduce the thermodynamic and vapor-liquid equilibrium properties [51, 52, 53], additionally, the force fields are compatible with the CVFF [46]. The LJ cross-interactions were given by the Lorentz-Berthelot combining rules [11]:  $\varepsilon_{ij} = \sqrt{\varepsilon_{ii}\varepsilon_{jj}}$  and  $\sigma_{ij} = (\sigma_{ii} + \sigma_{jj})/2$ . For the values of the LJ parameters and the partial charges, we refer the reader to the original papers [46, 47, 48, 49, 50].

## 2.3 Grand Canonical Monte Carlo

In the first part of this study, we consider the equilibrium behavior between the bulk-like free gas found in the shale macropores and the adsorbed gas in the OM structure. The free gas is either pure methane or equimolar mixtures of methane and carbon dioxide. The equilibrium

between these two phases, i.e., the equivalence of the chemical potentials of the gas species was established using GCMC [11, 12]. In the GCMC simulations, we consider a porous material at constant volume (the OM structure with kerogen fragments, asphaltene/resin molecules, and other dissolved species) in equilibrium with a gas reservoir (the bulk methane phase or bulk methane/carbon dioxide mixture phase) at temperature,  $T$ , and pressure,  $P$ . The reservoir imposes the chemical potentials,  $\mu_i(T, P)$ , where their values are inputs to the GCMC simulation. We determined the value of  $\mu_i$  by the Widom’s insertion method [54] using isothermal-isobaric Monte Carlo (NPT MC) simulations of the bulk phase. For a given  $\mu_i$  and  $T$ , the amount adsorbed is then given by the ensemble average of the number of adsorbed gas molecules in the OM structure. The adsorption isotherm was traced by increasing the pressure, where  $\mu_i$  determined in the corresponding NPT MC was imposed in the GCMC simulations. Additional simulation details regarding the GCMC simulations are provided in the Supplementary Material.

## 2.4 Molecular Dynamics

NVT and NPT MD simulations [11, 12] were used to perform the step-wise compression-cooling protocol for generating the microporous OM matrix described in Section 2.1. Further, NVT MD simulations were used to evaluate the diffusion of the states along the adsorption isotherms, where the equilibrium amounts of the adsorbed gases were determined from the GCMC simulations. Additional simulation details regarding the MD simulations are given in the Supplementary Material.

The in-plane diffusion of the adsorbed gas molecules  $\text{CH}_4$  and  $\text{CO}_2$  was described by the self-diffusivity,  $D_\alpha$ ,  $\alpha \equiv (\text{CH}_4, \text{CO}_2, \dots)$ , which was calculated using the Einstein equation from the mean-square displacement (MSD) parallel to the slit  $xy$ -plane, i.e.,

$$\text{MSD}_{xy}^\alpha(t) \propto 4D_\alpha t \tag{2}$$

In Eq. (2),  $\text{MSD}_{xy}^\alpha(t) \equiv \text{MSD}_x^\alpha(t) + \text{MSD}_y^\alpha(t)$ , where  $\text{MSD}_x^\alpha(t)$  and  $\text{MSD}_y^\alpha(t)$  are the MSDs in the  $x$ - and  $y$ -directions at time  $t$ , respectively. The determination of the MSDs includes averaging over different time origins [11, 12]. To estimate the time interval for fitting the values of  $D_\alpha$  according to Eq. (2), we first evaluated the slope of the MSD

$$\beta(t) = \frac{d \ln [\text{MSD}_{xy}^\alpha(t)]}{d \ln t} \tag{3}$$

as a function of time  $t$ , and selected a time interval over which  $\beta(t) \simeq 1$ , which corresponds to normal (Fickian) diffusion [55, 56].

In addition to determining the overall self-diffusivity  $D_\alpha$ , we also evaluated  $D_\alpha$  limited to either the microporous matrix or the mesopore slit, as described in Refs. [55, 56, 57]. The overall MSD is related to the MSDs in the microporous matrix and the mesopore slit as

$$\text{MSD}_{xy}^\alpha(t) = x_\alpha^{\text{micro}} \text{MSD}_{xy,\text{micro}}^\alpha(t) + x_\alpha^{\text{meso}} \text{MSD}_{xy,\text{meso}}^\alpha(t) \quad (4)$$

where  $x_\alpha^{\text{micro}}$  and  $x_\alpha^{\text{meso}}$  are the compositions of species  $\alpha$  in the microporous matrix and the mesopore slit, respectively, and  $\text{MSD}_{xy,\text{micro}}^\alpha(t)$  and  $\text{MSD}_{xy,\text{meso}}^\alpha(t)$  are the MSDs of species  $\alpha$  in the microporous matrix and mesopore slit at  $t$ , respectively. For normal diffusion

$$D_\alpha = x_\alpha^{\text{micro}} D_\alpha^{\text{micro}} + x_\alpha^{\text{meso}} D_\alpha^{\text{meso}} \quad (5)$$

where  $D_\alpha^{\text{micro}}$  and  $D_\alpha^{\text{meso}}$  are the self-diffusivities in the microporous matrix and the mesopore slit, respectively, evaluated using the Einstein equation [Eq.(2)].

### 3 Results and Discussion

The results of the simulation study are presented next. In Section 3.1, the structures of the OM matrices have been characterized based upon the geometric pore size distribution, the accessible surface area, the pore limiting diameter, and the micro- and mesopore volumes. In Sections 3.2 and 3.3, the adsorption and diffusion results are presented, respectively. All of the adsorption and diffusion simulations were performed at temperatures,  $T = \{298, 323, 343\}$  K, and pressures,  $P = \{50, 100, 275\}$  bar. The temperatures 323 and 343 K together with these pressures represent typical reservoir conditions [58], while the behavior at 298 K corresponds to ambient conditions. The model critical temperature,  $T_c$ , and critical pressure,  $P_c$ , are  $(T_c, P_c) \simeq (191 \text{ K}, 46 \text{ bar})$  for methane and  $(T_c, P_c) \simeq (306 \text{ K}, 77 \text{ bar})$  for  $\text{CO}_2$  [51, 52], which compare rather well with the experimental values  $(T_c, P_c) \simeq (191 \text{ K}, 46 \text{ bar})$  and  $(T_c, P_c) \simeq (304 \text{ K}, 74 \text{ bar})$ , respectively [59].

#### 3.1 Organic-Matter Structure Characterization

Examples of the dual-porosity OM structures generated from the step-wise compression-cooling strategy are shown in Fig. 2. Visual inspection of the microporous matrix indicates a rather tortuous pore network in the kerogen skeleton of the mature OM structures (Fig. 2a), and a parallel-stacked arrangement in the kerogen skeleton of the overmature OM structures (Fig. 2b). Further, we systematically characterized the OM structures by calculating the geometric pore size distribution (PSD) for the kerogen skeleton, the accessible surface area,

$S_A$ , and the pore limiting diameter,  $D_{\min}$ , of the kerogen in the microporous matrix. Details of these calculations are given next.

The PSD was determined by the following Monte Carlo procedure. A dense lattice of test points was placed in the simulation box and then the entire micropore space was probed by finding the largest spheres that contained test points, but did not overlap with the kerogen atoms. Each sphere then defined the volume of a void space that can be covered by a sphere of radius  $r$  or smaller,  $V_p(r)$ . The derivative  $-dV_p/dr$  is the PSD, which was obtained by numerical differentiation of  $V_p(r)$  [60, 61]. The PSDs for both of the OM structures are displayed in Fig. 2, where two distinct pore size regions are evident in all of the structures. The PSDs have a broad distribution of ultramicropores up to micropores, which are wider for the overmature OM structures (Fig. 2b) compared to the mature OM structures (Fig. 2a). Further, the PSDs exhibit a pronounced distribution of large micropores up to mesopores, the latter corresponding to the mesopore slit and its confining corrugated wall surfaces.

The accessible surface in the microporous matrix was obtained as a locus of the points that represents the location of the probe particle at a distance of the collision diameter  $\sigma$  from any atom in the microporous structure. A nitrogen atom with  $\sigma_N = 3.314 \text{ \AA}$  was used as a probe particle to directly relate the calculated  $S_A$  to that measured in BET adsorption experiments [3]. Application of the PoreBlazer tool [61] predicts  $S_A \simeq 125$  and  $175 \text{ m}^2/\text{g}$  for the mature and overmature kerogen in the microporous matrix. These values can be compared with  $S_A \simeq 95 \text{ m}^2/\text{g}$  for the mature type II OM structures by Collell et al. [31], and with  $S_A \simeq \{40, 75, 130, 150\} \text{ m}^2/\text{g}$  for the overmature type II kerogen structures by Michalec and Lísal [3], generated using LJ dummy particles with the size  $\sigma_{LJ} = \{9, 11, 13, 15\} \text{ \AA}$ .

The pore limiting diameter of the kerogen in the microporous matrix corresponds to a maximum probe size for which a pore network percolates. Application of the PoreBlazer tool [61] yields  $D_{\min} \simeq 7.8$  and  $11.3 \text{ \AA}$  for the mature and overmature kerogen skeletons, respectively, which indicates that normal diffusion will occur for both methane and carbon dioxide in the microporous matrix. It is worthwhile to mention that in the absence of the asphaltene/resin molecules and the other dissolved species that arise during the generation of the microporous matrix, values of  $D_{\min}$  would be rather small, and the microporous kerogen would not exhibit percolation for methane and carbon dioxide, unless dummy particles would be used [3, 26].

For methane adsorption in the OM structures (acting here as a proxy for shale gas adsorption), the excess adsorption amount was also computed, which requires determining the pore volume,  $V_{\text{pore}}$ . We approximated values of  $V_{\text{pore}}$  from the free volume obtained by the

following Monte Carlo procedure. Using the final equilibrated dual-porosity OM structures, we first performed NVT MD simulations for a few ns, periodically saving approximately a thousand configurations. Then for each of these configurations, we tested if a randomly inserted He-size probe particle with  $\sigma_{\text{He}} = 2.58 \text{ \AA}$  (viz helium pycnometry) overlapped with any atoms in the OM structure, repeating this overlap test  $10^6$  times. The fraction of insertions that does not overlap with any OM atom determines the free volume (as a fraction of the simulation box,  $V_{\text{box}}$ ) [62]. We determined the free volume and in turn  $V_{\text{pore}}$  separately in the microporous matrix and mesopore slit without considering the dissolved methane. The boundary between the microporous and mesopore regions was determined via the Gibbs dividing plane evaluated using the density profile of the kerogen skeleton, see Section 3.2. We obtained  $(V_{\text{pore}}^{\text{micro}}, V_{\text{pore}}^{\text{meso}}) \simeq (0.0276V_{\text{box}}, 0.149V_{\text{box}})$  and  $(V_{\text{pore}}^{\text{micro}}, V_{\text{pore}}^{\text{meso}}) \simeq (0.0244V_{\text{box}}, 0.248V_{\text{box}})$  for the mature OM structures with the 20  $\text{\AA}$  slit and 30  $\text{\AA}$  slit, respectively, and  $(V_{\text{pore}}^{\text{micro}}, V_{\text{pore}}^{\text{meso}}) \simeq (0.0460V_{\text{box}}, 0.127V_{\text{box}})$  and  $(V_{\text{pore}}^{\text{micro}}, V_{\text{pore}}^{\text{meso}}) \simeq (0.0416V_{\text{box}}, 0.207V_{\text{box}})$  for the overmature OM structures with the 20  $\text{\AA}$  slit and 30  $\text{\AA}$  slit, respectively.

## 3.2 Adsorption

### 3.2.1 Methane Adsorption

Fig. 3 presents adsorption isotherms for total and excess adsorption of methane in the OM structures studied. The  $\text{CH}_4$  total adsorption corresponds to the entire quantity of methane that resides in the OM structure, which includes both the methane within the micropores and mesopore slit (i.e., free gas), and the methane adsorbed directly on the micropore and mesopore surfaces (i.e., gas in an adsorbed state). The  $\text{CH}_4$  excess adsorption is the difference between the total adsorbed amount,  $n$ , and the amount of methane in the free volume of the OM micropores and mesopore slit, and characterizes the methane in the adsorbed state. The excess adsorption was estimated as

$$n^{\text{excess}} = n - (V_{\text{pore}}^{\text{micro}} + V_{\text{pore}}^{\text{meso}}) \rho_{TP} \quad (6)$$

where  $\rho_{TP}$  is the density of bulk methane at  $T$  and  $P$ , as obtained by NPT MC simulations.

From Fig. 3, several general trends are evident. First, the adsorption exhibits type-I Langmuir adsorption behavior, where the total adsorption increases with increasing pressure, which is the expected behavior for these materials. (The isotherm at  $T = 298 \text{ K}$  in the 20  $\text{\AA}$ -mature OM structure deviates from this pressure dependence, which is discussed below.) Second, the total adsorption decreases with increasing temperature, where the increase in the mobility (kinetic energy) of the methane molecules overcomes the attraction with the pore

surfaces resulting in less methane layered on the surface. As such, at higher temperature, less efficient packing of methane occurs as more free-gas methane occupies the mesopore volume. Third, the total adsorption increases with increasing size of the mesopore width, which is simply a consequence of an increase in the accessible pore volume. Lastly, the excess adsorption isotherms exhibit less dependence on the pressure, temperature and mesopore width, compared to the total adsorption isotherms. This indicates that the methane molecules prefer to occupy the porous volume, rather than forming layers on the surface.

A comparison of both the total and excess adsorption between the mature (Figs. 3a,c) and overmature (Figs. 3b,d) OM structures indicates stronger methane adsorption in the overmature OM structures, overall by about 50 %. This is primarily due to the larger accessible surface area in the microporous matrix for the overmature OM structures, which results in a relative increase in the excess adsorption amount.

As further insight, the adsorption behavior was evaluated separately in the microporous and mesopore regions of the OM structures. The microporous region corresponds to the microporous matrix, while the mesopore region corresponds to the mesopore slit and its confining corrugated surfaces. Fig. 4 provides examples of these quantities at  $T = 298$  and  $343$  K in the  $20 \text{ \AA}$ -mature and  $20 \text{ \AA}$ -overmature OM structures, where a complete set is provided in Figs. S1 to S4 of the Supplementary Material. Fig. 4 along with Figs. S1 to S4 indicate that the adsorption isotherms in the microporous region have minimal dependence with increasing  $P$ , where differences between the total and excess adsorbed amount are rather small with only a weak enhancement upon increasing  $P$ . This suggests that the microporous regions are nearly filled with methane for the  $P$ s considered in this study, and accessible adsorption sites are scarce. On the other hand, the total adsorption isotherms, and to a certain extent the excess adsorption isotherms in the mesopore region, exhibit a rather strong dependence with increasing  $P$ , where the differences between the total and excess adsorbed amount increase with increasing  $P$ . This indicates that methane molecules continue to both fill the mesopore space and adsorb on the mesopore surfaces, which still contain accessible adsorption sites. The assessments of the methane adsorption behavior in Fig. 4 and Figs. S1 to S4 are complemented by Figs. S5 to S8 of the Supplementary Material, which presents the fluid density profiles in the OM structures.

In contrast to the other cases considered in this study, the adsorption behavior at  $T = 298$  K in the  $20 \text{ \AA}$ -mature OM structure (Figs. 3a and 4a) exhibits a different pressure dependence, as characterized by a peak at  $P \sim 60$  bar in both the overall excess adsorption isotherm and the excess adsorption isotherm in the mesopore region. For this particular temperature

and particular mesopore width, this suggests that the accessible surface in the mesopore region has been saturated with methane molecules at  $P \sim 60$  bar. Above this  $P$ , adsorption only occurs as free gas above the adsorbed layers. Therefore, as this amount increases with increasing  $P$ , then relative to the total amount adsorbed, the excess adsorption amount decreases. Note that the excess adsorption isotherm in the microporous region is independent of  $P$ , which indicates that the adsorbed layers have been saturated in the microporous region at  $P < 60$  bar. The fluid density profiles in the 20 Å-mature OM structure at  $T = 298$  K shown in Fig. S9 of the Supplementary Material, further supports this assessment. For the higher temperatures and other OM structures studied here, we expect that the excess adsorption isotherms will exhibit a peak located at  $P > 275$  bar.

### 3.2.2 Competitive Adsorption of Methane and Carbon Dioxide Mixtures

Fig. 5 presents the adsorption isotherms of the equimolar  $\text{CH}_4/\text{CO}_2$  mixtures in the 20 Å-mature and 20 Å-overmature OM structures, along with the corresponding selectivity parameter,  $S_{\text{CO}_2}$ . Fig. S10 in the Supplementary Material displays the analogous results in the 30 Å-mature and 30 Å-overmature OM structures. The selectivity parameter characterizes adsorption selectivity as the ratio of the mole fractions of the two species in the adsorbent relative to the ratio of the mole fractions in the bulk phase. As such,  $S_{\text{CO}_2}$ , defined as

$$S_{\text{CO}_2} = \frac{x_{\text{CO}_2}/x_{\text{CH}_4}}{y_{\text{CO}_2}/y_{\text{CH}_4}} \quad (7)$$

is a criterion to assess the relative adsorption preference between  $\text{CH}_4$  and  $\text{CO}_2$  in a binary mixture. In Eq. (7),  $x_{\text{CO}_2}$  and  $x_{\text{CH}_4}$  are the average mole fractions of  $\text{CO}_2$  and  $\text{CH}_4$  in the adsorbent, while  $y_{\text{CO}_2}$  and  $y_{\text{CH}_4}$  are the mole fractions of  $\text{CO}_2$  and  $\text{CH}_4$  in the bulk phase. A larger adsorption selectivity value,  $S_{\text{CO}_2}$ , corresponds to a greater preference of  $\text{CO}_2$  to adsorb in the porous material compared to  $\text{CH}_4$ .

The adsorption isotherms in Figs. 5a,b and S10a,b indicate that  $\text{CO}_2$  is preferentially adsorbed over  $\text{CH}_4$  in all of the OM structures, and at all  $T$ s and  $P$ s studied. The  $\text{CH}_4$  adsorption isotherms increase with increasing  $P$ , and decrease with increasing  $T$ , which is analogous to the pure methane adsorption behavior. Also analogous to the pure methane adsorption behavior,  $\text{CH}_4$  adsorption is stronger in the overmature OM structures compared to the mature OM structures. However, in contrast to the pure methane adsorption behavior, the  $\text{CH}_4$  adsorption only slightly increases with increasing size of the mesopore width. The  $\text{CO}_2$  adsorption isotherms at  $T = 298$  K, and to a certain extent those at  $T = 323$  K, sharply increase between 50 to 100 bar, which corresponds to the  $P_c$  of  $\text{CO}_2$ . At higher pressures, the

isotherms exhibit a rather weak dependence. On the contrary, the CO<sub>2</sub> adsorption isotherms at  $T = 343$  K gradually increase with increasing  $P$ . The sharp increase in the CO<sub>2</sub> adsorption is associated with a sharp increase in the bulk density around  $P_c$  of CO<sub>2</sub>, as evident in Fig. S11 of the Supplementary Material. Fig. S11 displays the bulk isotherms for CO<sub>2</sub> (and also CH<sub>4</sub> for the sake of comparison) at all  $T$ s studied, as obtained by our NPT MC simulations. Similar to methane adsorption, the CO<sub>2</sub> adsorption isotherms at  $P > 50$  bar decrease with increasing  $T$ , and exhibit a stronger CO<sub>2</sub> adsorption in the overmature OM structures compared to the mature OM structures, overall by about 50 %.

The pressure dependence of  $S_{\text{CO}_2}$ , displayed in Figs. 5c,d and S10c,d, further emphasizes the preferential adsorption of CO<sub>2</sub> over CH<sub>4</sub>. The values of  $S_{\text{CO}_2}$  range between 2.0-4.0, where the overmature OM structures display higher CO<sub>2</sub> adsorption selectivity compared to the mature OM structures. Our simulated values of  $S_{\text{CO}_2}$  are within the reported experimental values of 1.9-8.9 [63, 64]. The sharp increase in the CO<sub>2</sub> adsorption isotherms at  $T = 298$  and 323 K between 50-100 bar coincides with a corresponding sharp increase in  $S_{\text{CO}_2}$  with increasing  $P$ , peaking around the  $P_c$  of CO<sub>2</sub>; at higher  $P$ ,  $S_{\text{CO}_2}$  gradually decreases. In contrast, the CO<sub>2</sub> adsorption selectivity at  $T = 343$  K gradually decreases with increasing  $P$ . Above the  $P_c$  of CO<sub>2</sub>, the CO<sub>2</sub> adsorption selectivity decreases with increasing  $T$ , which indicates that the CO<sub>2</sub> adsorption decreases faster than the CH<sub>4</sub> adsorption upon increasing  $T$ . However above 200 bar,  $S_{\text{CO}_2}$  appears to be nearly independent of  $T$ .

Fig. 6 and Figs. S12 to S15 of the Supplementary Material detail the preferential adsorption of CO<sub>2</sub> over CH<sub>4</sub> by presenting the adsorption isotherms of the equimolar CH<sub>4</sub>/CO<sub>2</sub> mixtures separately in the microporous and mesopore regions of the OM structures. The figures indicate that CO<sub>2</sub> is preferentially adsorbed over CH<sub>4</sub> in both the microporous and mesopore regions. A comparison of the CH<sub>4</sub> and CO<sub>2</sub> adsorption isotherms in the 20 Å-mature and 20 Å-overmature OM structures (Figs. 6, S12 and S13) indicates CH<sub>4</sub> more strongly adsorbs in the mesopore region than in the microporous region in both OM structures, while adsorption in the microporous region marginally increases with  $P$ . CO<sub>2</sub> exhibits similar behavior in the 20 Å-mature OM structure. However, in contrast, CO<sub>2</sub> adsorption becomes stronger in the microporous region than in the mesopore region for the 20 Å-overmature OM structure, while CO<sub>2</sub> adsorption in both regions decreases with increasing  $P$ . Moreover, as the mesopore width increases to 30 Å (Figs. S14 and 15), adsorption of both CH<sub>4</sub> and CO<sub>2</sub> in the mesopore region increases for the 30 Å-overmature OM structure, while CO<sub>2</sub> adsorption in both the microporous and mesopore regions becomes comparable.

The CH<sub>4</sub> and CO<sub>2</sub> adsorption behavior in the microporous and mesopore regions of the

OM structures is complemented by computing the corresponding values of  $S_{\text{CO}_2}$ , presented in Figs. S16 to S19 of the Supplementary Material. Values of  $S_{\text{CO}_2}$  are about 2 in the mesopore regions, while values of nearly 6 are observed in the microporous regions, suggesting that both microporous OM matrices are strong adsorbents for carbon dioxide. Finally, similar to pure methane adsorption, we present fluid density profiles in the OM structures (Figs. S20 to S23 of the Supplementary Material) to further emphasize the preferential adsorption of  $\text{CO}_2$  over  $\text{CH}_4$  in the OM structures studied.

### 3.3 Molecular Diffusion

The diffusion of methane molecules, and the competing diffusion of methane and carbon dioxide in OM is critical, respectively, in shale gas transport and enhanced shale gas recovery by carbon dioxide. Generally, confinement hinders the diffusion of the adsorbed molecules inside the micro-, as well as mesopores, and restricts access to the adsorption sites. Diffusion in the porous material is affected by the porous structure (size of pores, pore connectivity and tortuosity), and the affinity of the adsorbates towards the pore surfaces.

#### 3.3.1 Methane Self-Diffusivity

Fig. 7 displays the self-diffusivity of methane,  $D_{\text{CH}_4}$ , along with the  $\text{CH}_4$  adsorption isotherms for all of the OM structures studied. Four general trends are observed. First,  $D_{\text{CH}_4}$  decreases as  $P$  increases due to the increase of  $\text{CH}_4$  adsorption with increasing  $P$ . Second,  $D_{\text{CH}_4}$  increases with increasing  $T$  due to the increase in the mobility (kinetic energy) of the adsorbed  $\text{CH}_4$  molecules with increasing  $T$ . Third, due to the stronger  $\text{CH}_4$  adsorption in the overmature OM structures compared to the mature OM structures,  $D_{\text{CH}_4}$  is lower in the overmature OM structures by about 20 %. Finally,  $D_{\text{CH}_4}$  increases with increasing mesopore width because the  $\text{CH}_4$  molecules adsorbed as free gas have a higher mobility in the wider mesopores compared to the narrower mesopores.  $D_{\text{CH}_4}$  along the adsorption isotherm at  $T = 298$  K for the 20 Å-mature OM exhibits a moderately different pattern due to the distinct adsorption behavior discussed above.

To provide further insight into  $\text{CH}_4$  diffusion in the OM structures, Fig. 8 presents examples of  $D_{\text{CH}_4}$  along the adsorption isotherms at  $T = 298$  and 343 K in the 30 Å-mature and 30 Å-overmature OM structures, evaluated separately in the microporous and mesopore regions. A complete set of  $D_{\text{CH}_4}$  along all  $\text{CH}_4$  adsorption isotherms is provided in Figs. S24 to S27 of the Supplementary Material. Fig. 8, together with Figs. S24 to S27, as expected, indicate that  $D_{\text{CH}_4}$  in the mesopore region is enhanced with respect to the overall  $D_{\text{CH}_4}$ , while  $D_{\text{CH}_4}$

in the microporous region is reduced with respect to the overall  $D_{\text{CH}_4}$ . Furthermore,  $D_{\text{CH}_4}$  in both regions decreases with increasing  $P$ , while increases with increasing  $T$ , similar to the overall  $D_{\text{CH}_4}$ . The enhancement of  $D_{\text{CH}_4}$  in the mesopore region is about 20 % in both OM structures, which is due to the higher mobility of  $\text{CH}_4$  in the centre of the mesopores, where  $\text{CH}_4$  molecules behave as free gas. On the other hand, the reduction of  $D_{\text{CH}_4}$  in the microporous region is more pronounced for the mature OM structures (by about 50 %) than for the overmature OM structures (by about 20 %). Moreover,  $D_{\text{CH}_4}$  in the microporous region of the overmature OM structures becomes higher than or comparable with  $D_{\text{CH}_4}$  in the microporous region of the mature OM structures despite the stronger  $\text{CH}_4$  adsorption in the microporous overmature OM matrices. This behavior is associated with the parallel-stacked arrangement of the kerogen fragments in the microporous overmature OM matrices, which enhances diffusion when compared with the rather tortuous pore kerogen network in the microporous mature OM matrices.

### 3.3.2 Self-Diffusivity of Methane and Carbon Dioxide Mixtures

Enhanced shale gas recovery by carbon dioxide injection relies on: (i) the preferential adsorption of  $\text{CO}_2$  compared to the shale gas in the shale rock OM; and (ii) sufficient diffusion of the adsorbed  $\text{CO}_2$  molecules within the shale rock OM. In Fig. 9, we display the  $\text{CH}_4$  and  $\text{CO}_2$  self-diffusivity,  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  respectively, along the mixture adsorption isotherms for all of the OM structures studied. It is evident that due to the preferential adsorption of  $\text{CO}_2$  compared with  $\text{CH}_4$ ,  $D_{\text{CO}_2}$  is lower than  $D_{\text{CH}_4}$  by a factor of 3-4 at  $P < 100$  bar, and by a factor of about 2 at the higher  $P$ . Both  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  decrease with increasing  $P$ , and increase with increasing  $T$  and increasing mesopore width. Note that  $D_{\text{CH}_4}$  decreases rather rapidly at  $P < 100$  bar, which correlates with the sharp increase in the  $\text{CO}_2$  adsorption around the  $P_c$  of  $\text{CO}_2$  (cf. Figs. 5a,b and S10a,b). Both  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  in the overmature OM structures are only slightly lower than those in the mature OM structures (by about 10 %), despite both  $\text{CH}_4$  and  $\text{CO}_2$  adsorbing more strongly in the overmature OM structures compared to the mature OM structures (by about 50 %). Similar to the  $\text{CH}_4$  and  $\text{CO}_2$  adsorption behavior, this behavior is due to the distinctively different structural characteristics of the porous networks (i.e., the parallel-stacked arrangement of the kerogen fragments for the overmature OM structures compared to the rather tortuous pore kerogen network of the mature OM structures).

To further detail the diffusion of the adsorbed  $\text{CH}_4$  and  $\text{CO}_2$  molecules in the OM structures, Fig. 10 presents examples of  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  along the adsorption isotherms at  $T = 298$  and 343 K in the 30 Å-mature and 30 Å-overmature OM structures, evaluated separately

in the microporous and mesopore regions. A complete set of  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  along all the adsorption isotherms is provided in Figs. S28 to S31 of the Supplementary Material. It is evident that in the mesopore regions, both  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  are enhanced with respect to the overall  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  by 20 to 50 %. In the micropore regions,  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  are reduced with respect to the overall  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  by a factor of 2-3 for the mature OM structures, and by a factor slightly less than 1 for the overmature OM structures. In both regions,  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  decrease with increasing  $P$ , and increase with increasing  $T$ , similar to the overall  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$ .

## 4 Conclusions

We employed molecular dynamics (MD) simulations and generated realistic models of organic matter (OM) structures with both micro- and mesoporosity, which correspond to mature and overmature type-II kerogens. While most previous simulation studies have modeled only the kerogens in the OM structures, in contrast, the model OM structures in this study contain asphaltene/resin, hydrocarbons and carbon dioxide/water fractions. The result of including these other components was a realistic pore network percolation allowing for the diffusion of gas molecules within the microporous matrix of the OM structures. The microporous matrix of the mature OM structures exhibited a rather tortuous pore network in the kerogen skeleton, while the overmature OM structures displayed a parallel-stacked arrangement of kerogen fragments.

To provide molecular-level insight into shale gas adsorption and transport in shale rocks, we evaluated the adsorption and self-diffusivity of methane as a proxy of shale gas using grand canonical Monte Carlo (GCMC) and MD simulations over temperatures and pressures corresponding to typical reservoir conditions. For pure methane in the OM structures, the main findings from our work are:

- The adsorption isotherms exhibit type-I Langmuir adsorption behavior. The total adsorption, and to a certain extent the excess adsorption, increases with increasing pressure as the pressure drives the methane molecules into both the OM micro- and mesopores. Both the total and excess adsorption decreases with increasing temperature due to the increased mobility of the methane molecules, which prefer to behave as free gas in the mesopore rather than adsorb to the pore surfaces.
- The adsorption is higher in the overmature OM structures compared to the mature OM structures, which correlates with the larger accessible surface area of the microporous matrix

in the overmature OM structures. This behavior is also due to the parallel stacking of the kerogen fragments in the overmature OM structures. The latter promotes stronger methane-kerogen surface interactions compared with the tortuous pore network in the mature OM structures.

- The self-diffusivity,  $D_{\text{CH}_4}$ , along the adsorption isotherms anti-correlates with the adsorption behavior, i.e.,  $D_{\text{CH}_4}$  decreases with pressure, increases with temperature, and is lower in the overmature OM structures than in the mature OM structures.
- Values of  $D_{\text{CH}_4}$  are reduced in the OM micropores by a factor of about 2 for the mature OM structures, and by a factor of less than 1 for the overmature OM structures, which is associated with the structural characteristics of pore network. The parallel stacking of the kerogen fragments in the overmature OM structures enhances methane diffusion as opposed to the tortuous pore network in the mature OM structures.

To provide molecular-level insight into the displacement of shale gas by carbon dioxide during enhanced shale gas recovery, we further investigated adsorption and self-diffusivity of equimolar methane/carbon dioxide mixtures by GCMC and MD simulations at the relevant reservoir temperatures and pressures. For the equimolar methane/carbon dioxide mixtures in the OM structures, the main findings from our work are:

- $\text{CO}_2$  preferentially adsorbs compared with methane in both of the OM structures studied with a  $\text{CO}_2$  selectivity in the range of 2-4. Both the preferential adsorption and selectivity of  $\text{CO}_2$  are more pronounced in the overmature OM structures compared to the mature OM structures, which correlates with both the larger accessible surface area of the microporous matrix and the parallel stacking of the kerogen fragments in the overmature OM structures.
- The  $\text{CH}_4$  adsorption behavior in the mixture is analogous to the pure methane adsorption behavior. The  $\text{CO}_2$  adsorption isotherms are affected by a sharp increase in the bulk density near the  $\text{CO}_2$  critical pressure, where the  $\text{CO}_2$  adsorption isotherms at temperatures near the  $\text{CO}_2$  critical temperature sharply increase, reaching a maximum near the  $\text{CO}_2$  critical pressure. Above the  $\text{CO}_2$  critical pressure, the  $\text{CO}_2$  adsorption isotherms, depending on temperature, either slightly increase or slightly decrease with increasing pressure.
- The  $\text{CO}_2$  separation factors at the lower temperatures also exhibit maximums near the  $\text{CO}_2$  critical pressure, and then decrease with increasing pressure. The  $\text{CO}_2$  separation factors further decrease with increasing temperature, exhibiting faster decrease in the  $\text{CO}_2$  adsorption compared with the  $\text{CH}_4$  adsorption upon increasing temperature. The  $\text{CO}_2$  separation

is significantly enhanced in the micropores with respect to the mesopores, where the CO<sub>2</sub> separation factors may reach values of up to 6.

- Self-diffusivities,  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$ , along the mixture adsorption isotherms, in general anti-correlate with the mixture adsorption behavior, i.e.,  $D_{\text{CO}_2}$  is lower than  $D_{\text{CH}_4}$ , depending on pressure, by a factor of 2-4. Both  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  decrease with increasing pressure, and increase with increasing temperature. Despite stronger CH<sub>4</sub> and CO<sub>2</sub> adsorption in the overmature OM structures,  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  are either only slightly lower than or comparable with those in the mature OM structures. This behavior is due to the enhancement of molecular diffusion by the parallel stacking of the kerogen fragments in the overmature OM structures.
- Values of  $D_{\text{CH}_4}$  and  $D_{\text{CO}_2}$  are reduced by a factor of about 3 and by factor of about 1 in the microporous matrix of the mature and overmature OM structures, respectively, reflecting the distinct differences in the structural characteristics of the pore network.

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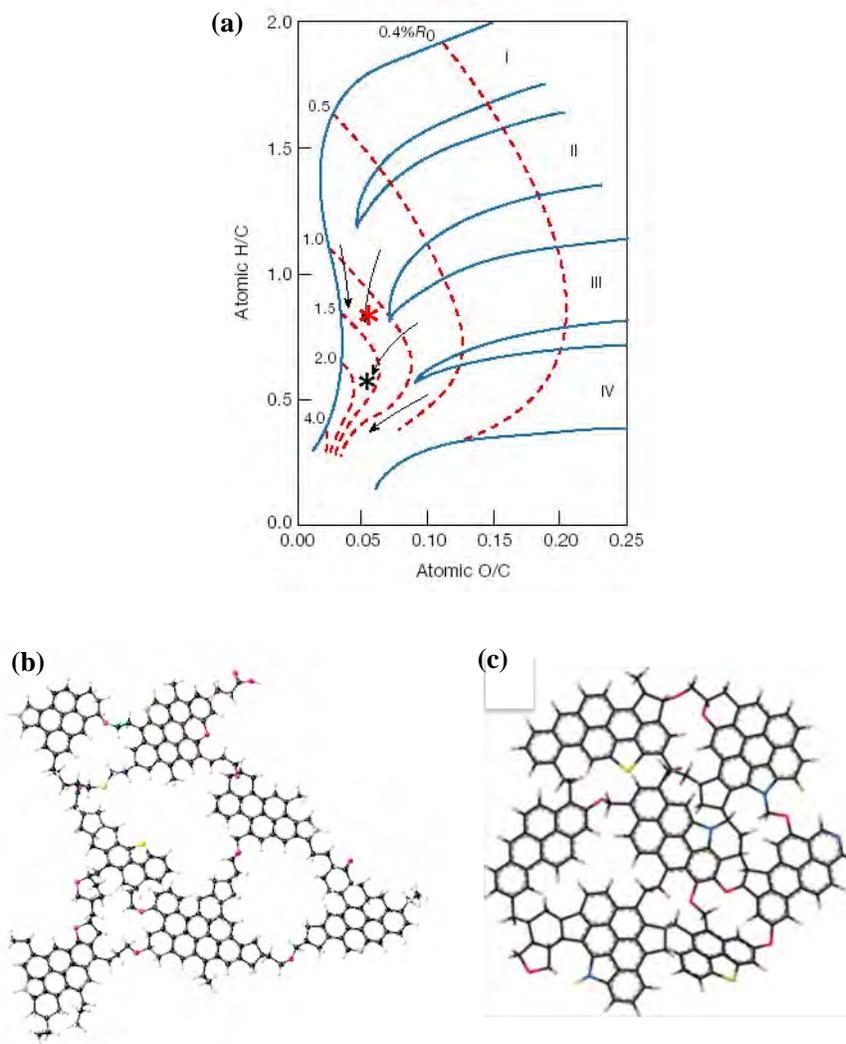


Figure 1: (a) The van Krevelen diagram;  $R_0$  is the vitrinite reflectance [14, 15]. The red and black stars indicate, respectively, the mature and overmature type II kerogens studied in this work. (b) Molecular model of a kerogen unit representing mature type II kerogen. The chemical formula of the kerogen unit is  $C_{242}H_{219}O_{13}S_2N_5$ , its molecular weight  $M_w = 3468.4$  g/mol, and the atomic ratios are  $H/C=0.905$ ,  $O/C=0.054$ ,  $S/C=0.008$  and  $N/C=0.021$ . (c) Molecular model of a kerogen unit representing overmature type II kerogen. The chemical formula of the kerogen unit is  $C_{175}H_{102}O_9S_2N_4$ , its molecular weight  $M_w = 2468.9$  g/mol, and the atomic ratios are  $H/C=0.58$ ,  $O/C=0.051$ ,  $S/C=0.011$  and  $N/C=0.023$ . The red, blue, yellow, grey and white spheres represent oxygen, nitrogen, sulphur, carbon and hydrogen atoms, respectively. Alternative composition and structural parameters of the kerogen units can be found in Ref. [24].

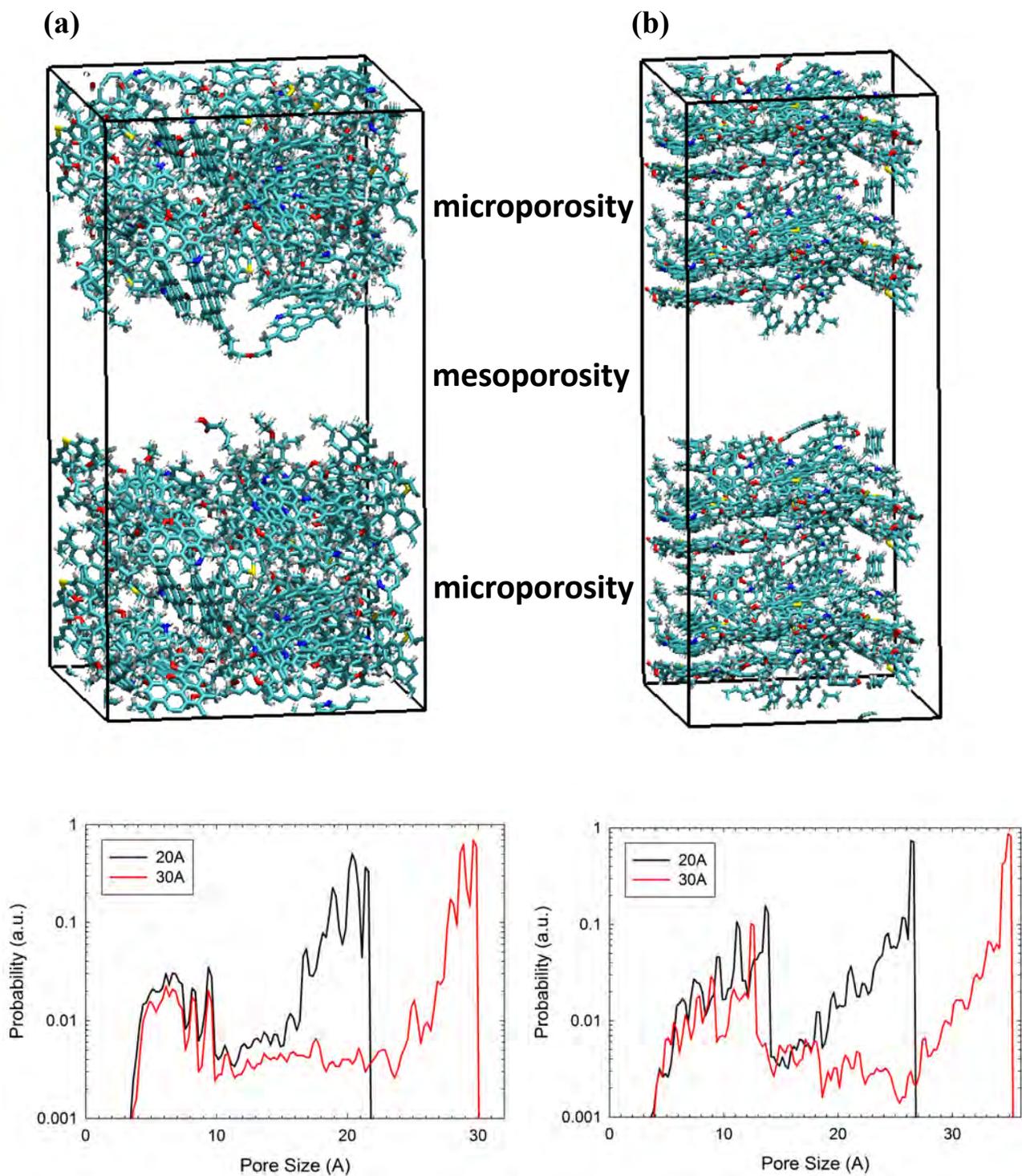


Figure 2: Configuration snapshots (top) and pore size distributions (bottom) for the (a) mature and (b) overmature organic matter structures. For visual clarity, only the kerogen matrix is shown in each snapshot.

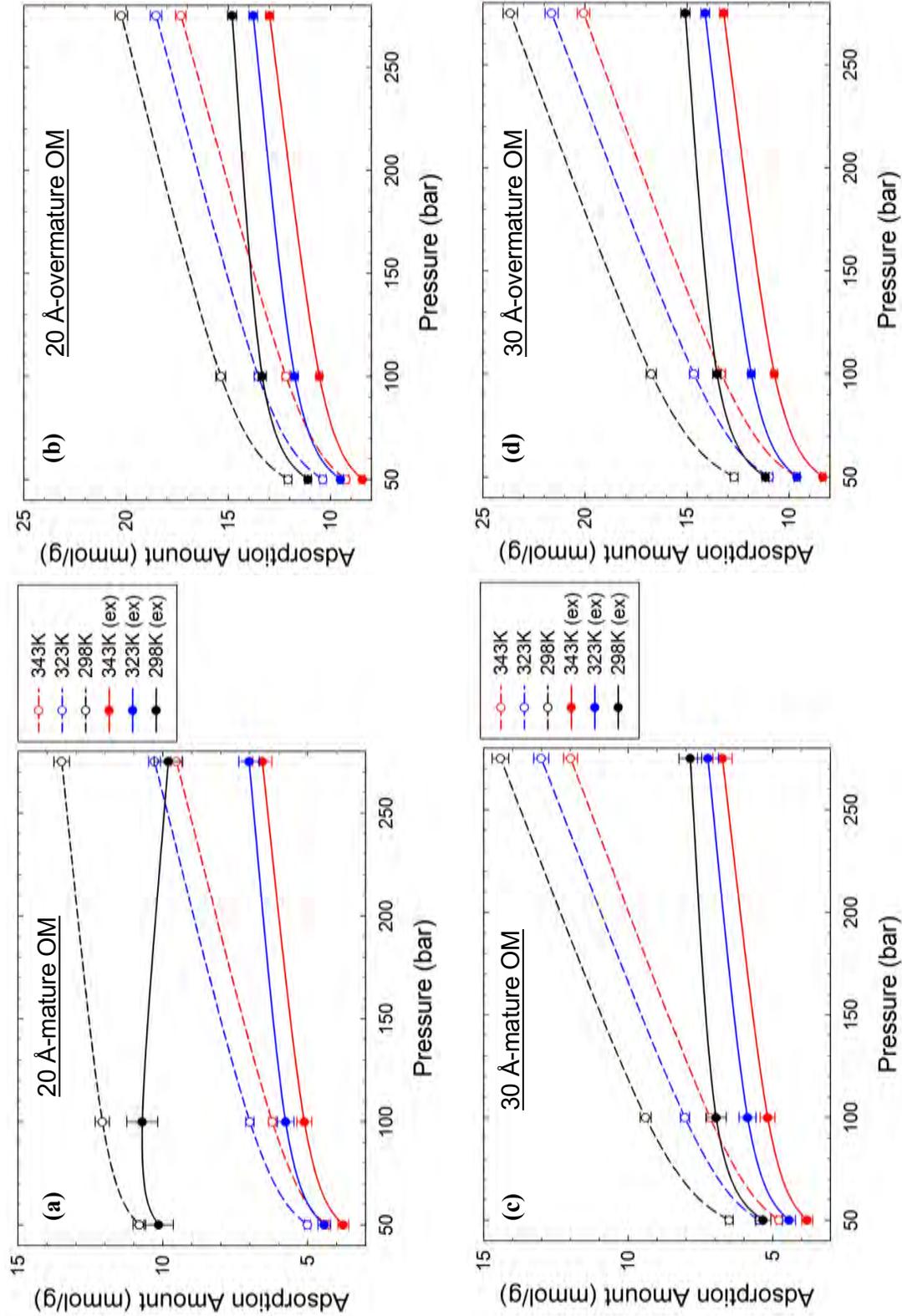


Figure 3: Adsorption isotherms of methane in the mature and overmature organic matter structures. The dashed lines and open symbols denote the total adsorption amount, while the solid lines and filled symbols correspond to the excess adsorption amount. The lines serve as a guide to the eye only.

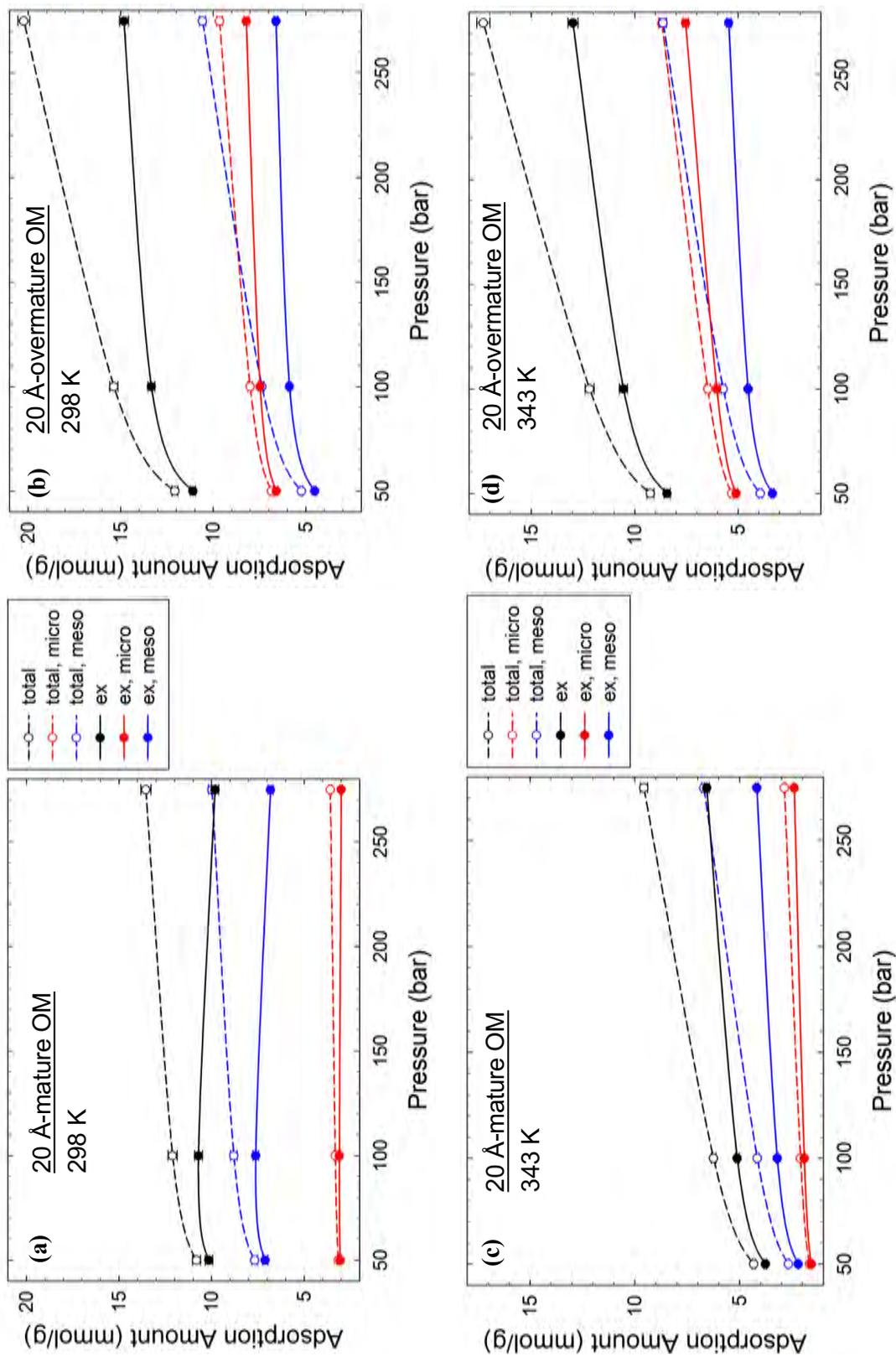


Figure 4: Adsorption isotherms of methane at temperatures of 298 and 343 K in the microporous and mesopore regions of the 20 Å-mature and 20 Å-overmature organic matter structures. The dashed lines and open symbols denote the total adsorption amount, while the solid lines and filled symbols correspond to the excess adsorption amount. The lines serve as a guide to the eye only.

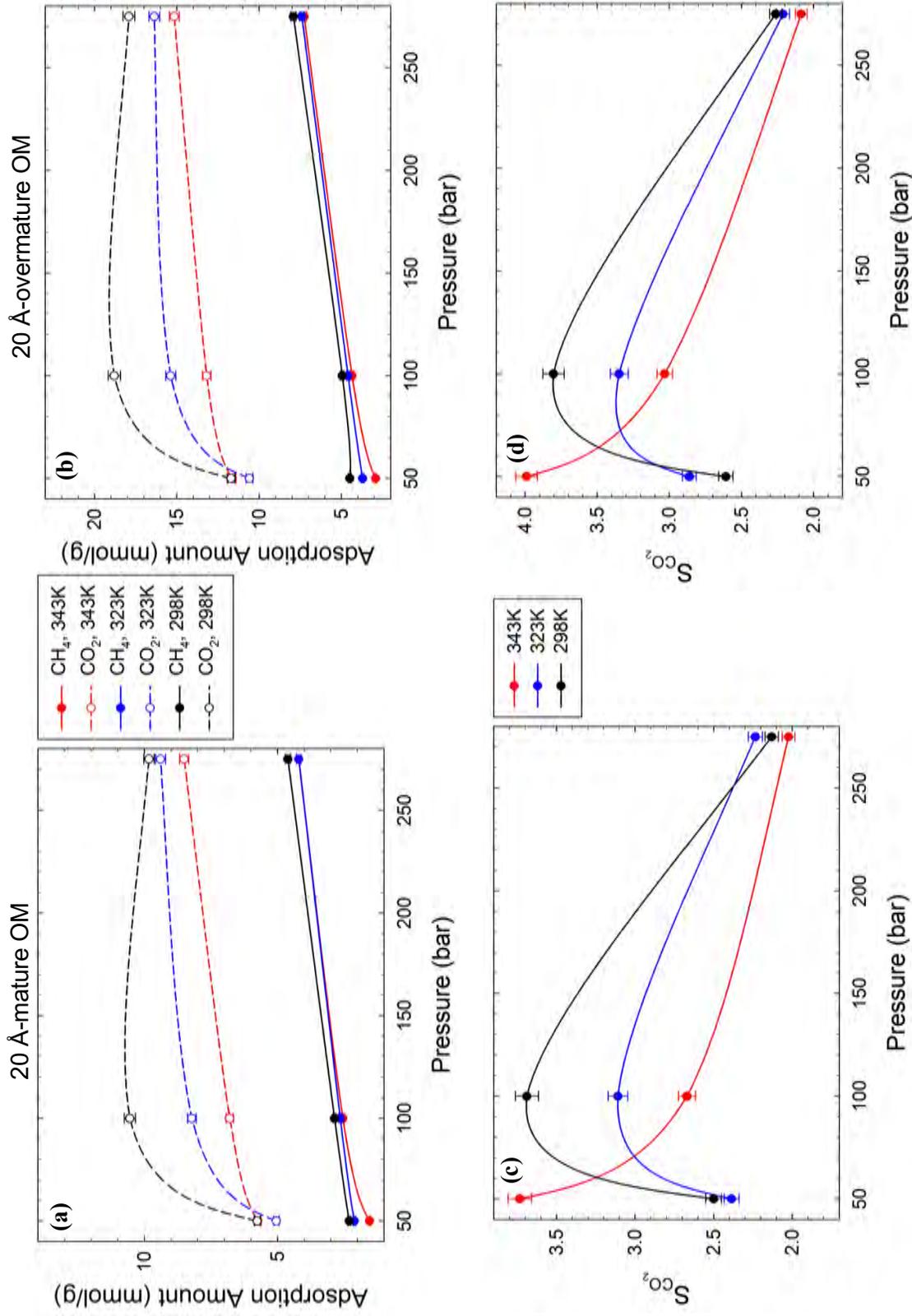


Figure 5: (a, b) Adsorption isotherms and (c, d) selectivity parameter,  $S_{CO_2}$ , of equimolar methane/carbon dioxide mixtures in the 20 Å-mature and 20 Å-overmature organic matter structures. The solid lines and filled symbols denote the results for methane, while the dashed lines and open symbols correspond to the results for carbon dioxide. The lines serve as a guide to the eye only.

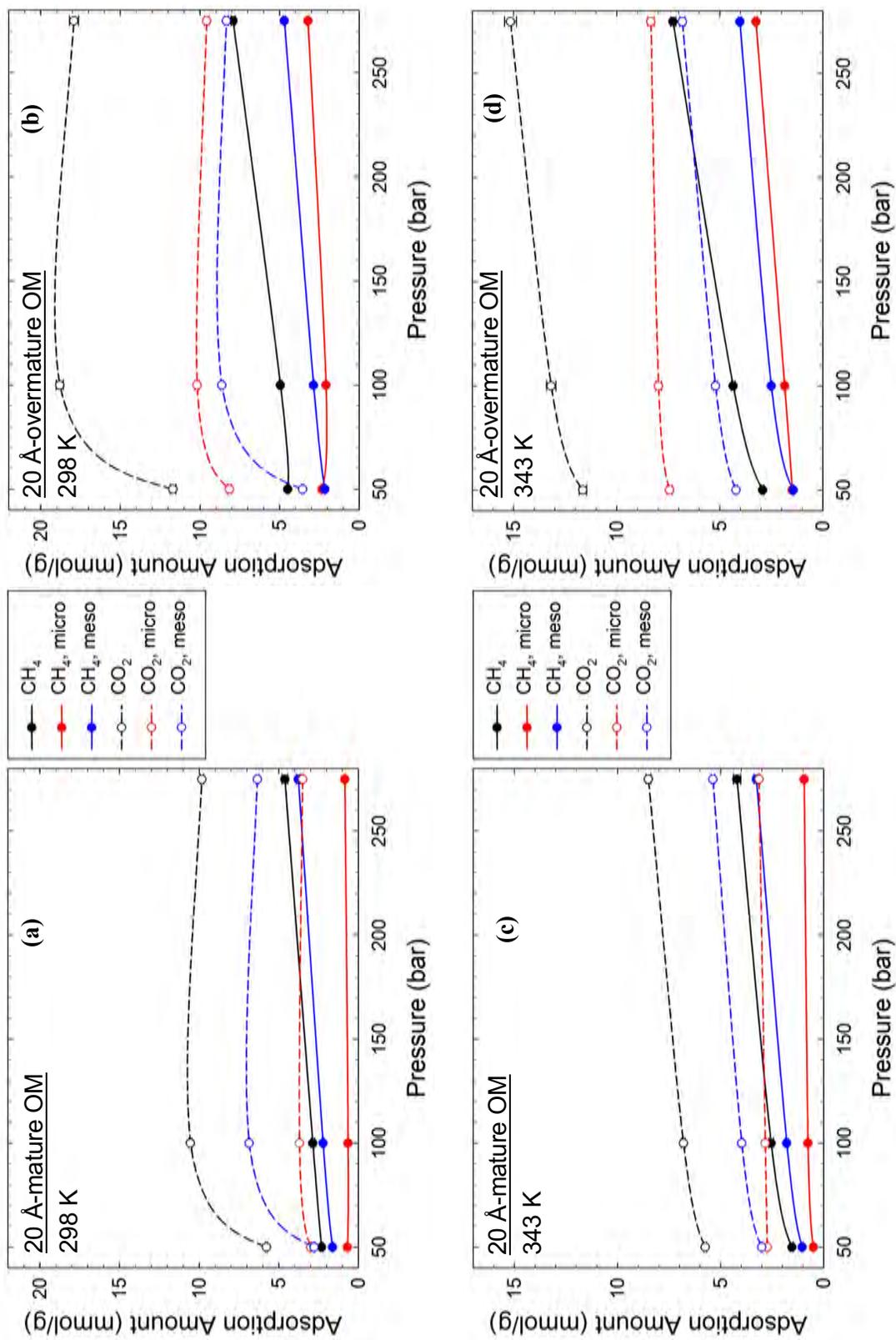


Figure 6: Adsorption isotherms of equimolar methane/carbon dioxide mixtures at temperatures of 298 and 343 K in the microporous and mesopore regions of the 20 Å-mature and 20 Å-overmature organic matter structures. The solid lines and filled symbols denote the results for methane, while the dashed lines and open symbols correspond to the results for carbon dioxide. The lines serve as a guide to the eye only.

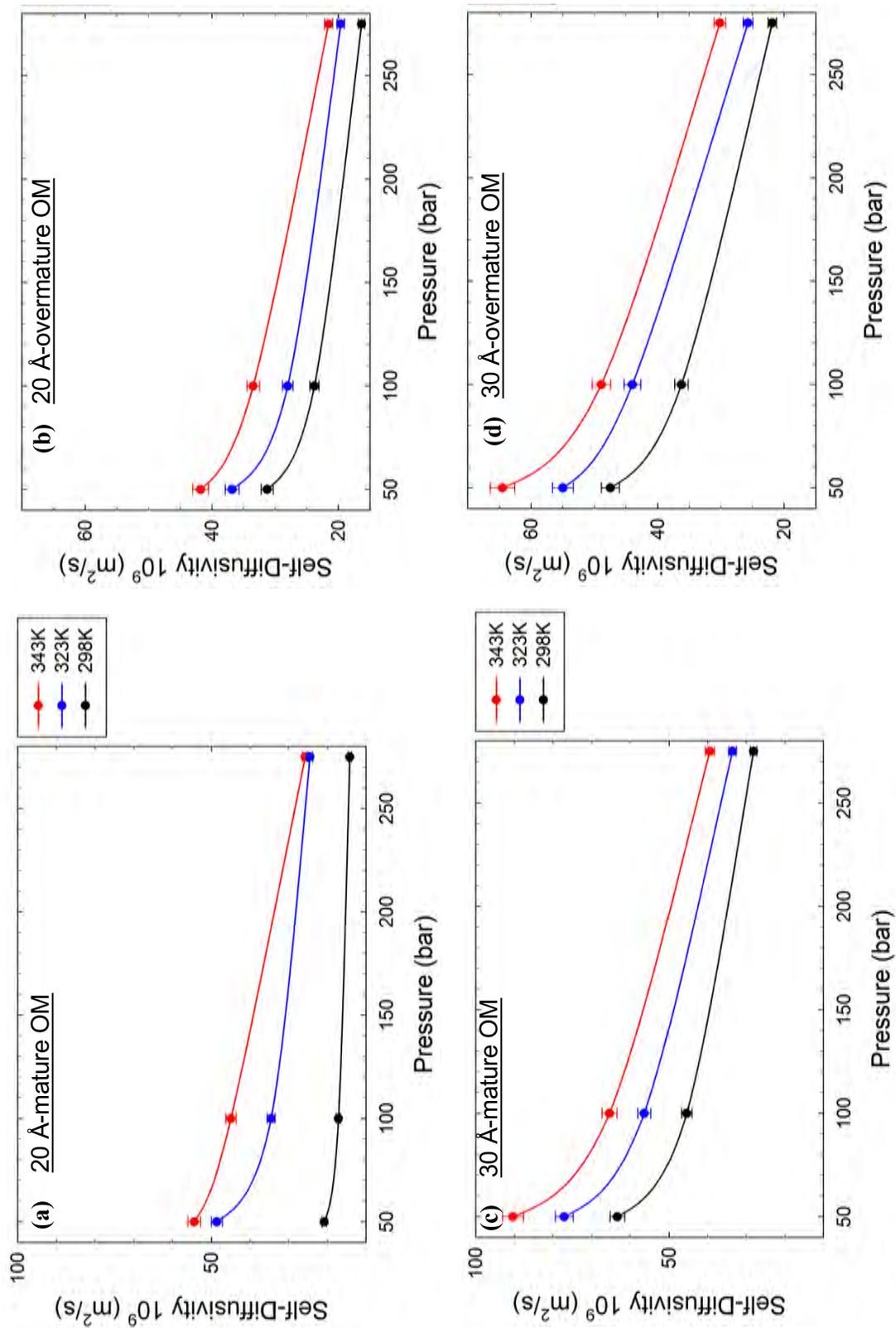


Figure 7: The self-diffusivity of methane along the adsorption isotherms in the mature and overmature organic matter structures. The lines serve as a guide to the eye only.

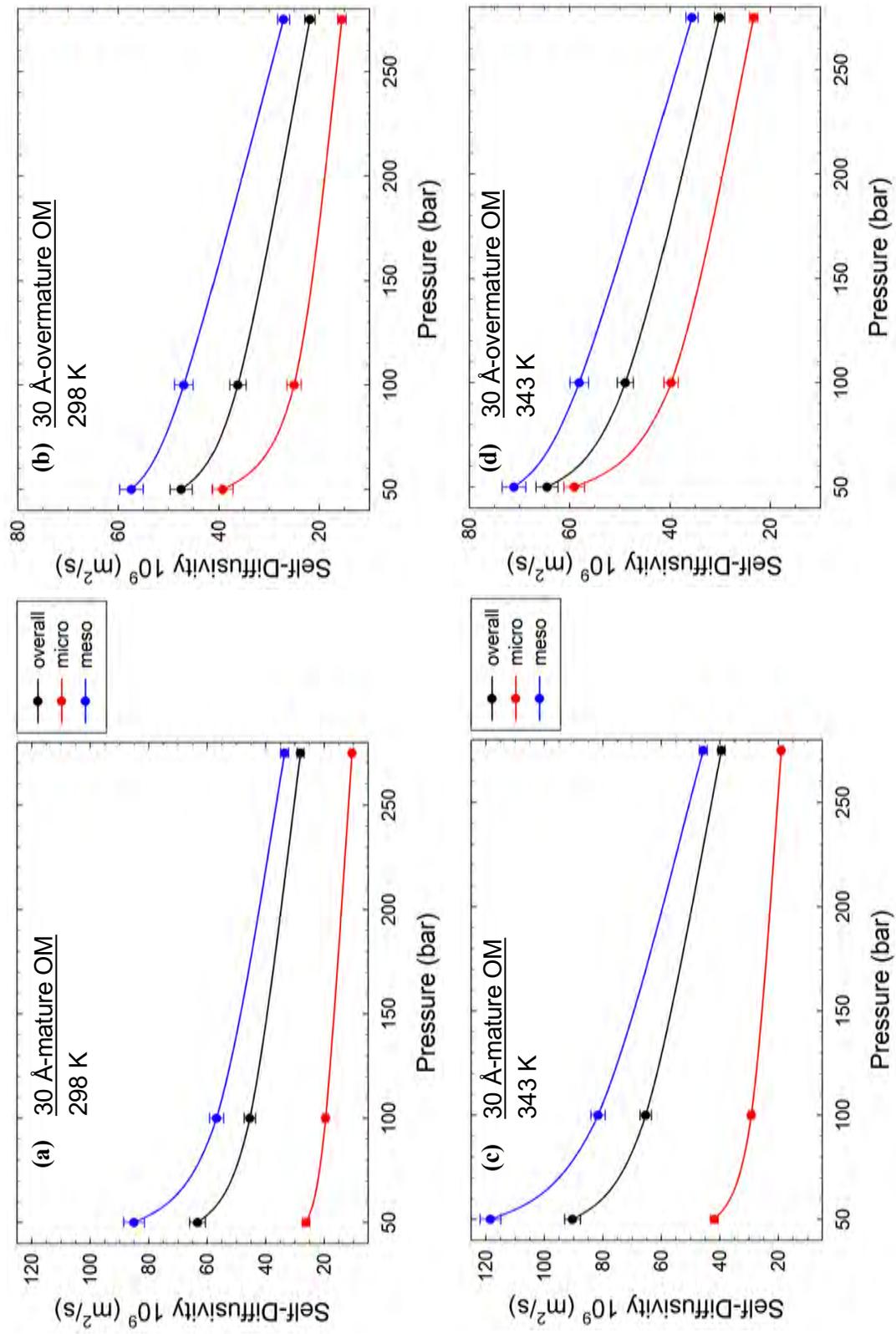


Figure 8: The self-diffusivity of methane along the adsorption isotherms at temperatures of 298 and 343 K in the microporous and mesopore regions of the 30 Å-mature and 30 Å-overmature organic matter structures. The lines serve as a guide to the eye only.

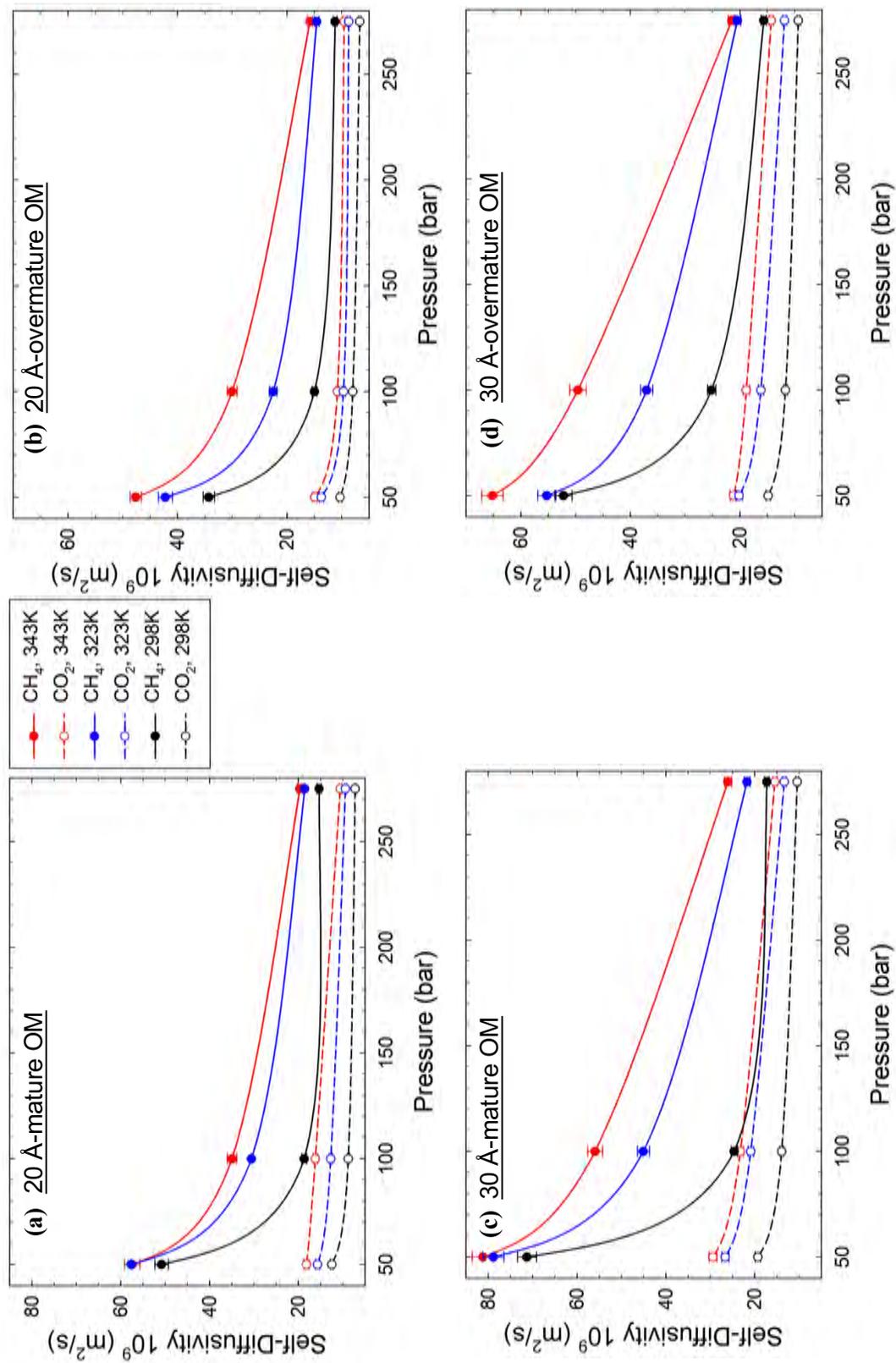


Figure 9: The self-diffusivity of methane and carbon dioxide along the mixture adsorption isotherms in the mature and over-mature organic matter structures. The solid lines and filled symbols denote the results for methane, while the dashed lines and open symbols correspond to the results for carbon dioxide. The lines serve as a guide to the eye only.

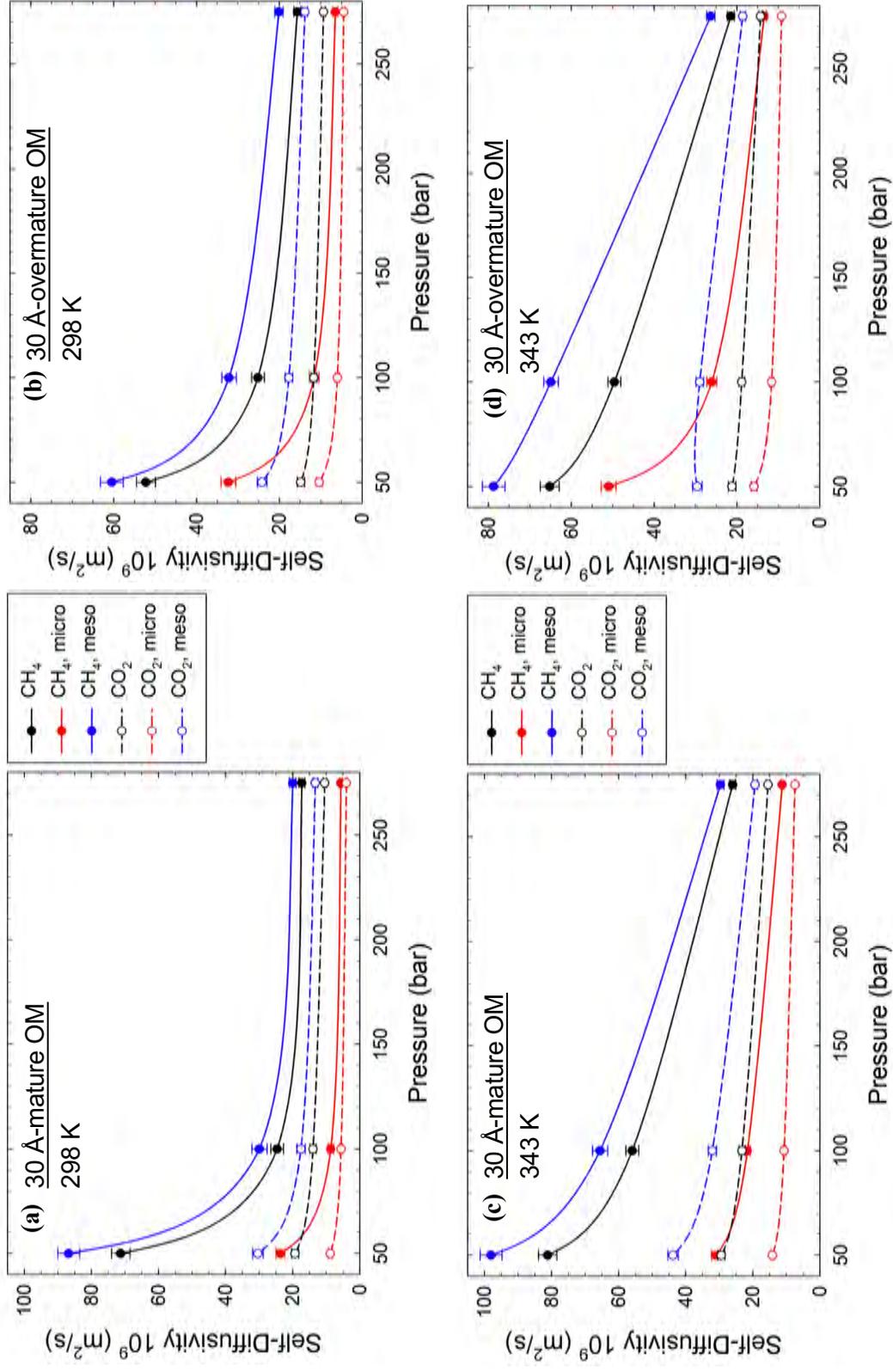


Figure 10: The self-diffusivity of methane and carbon dioxide along the mixture adsorption isotherms at temperatures of 298 and 343 K in the microporous and mesopore regions of the 30 Å-mature and the 30 Å-overmature organic matter structures. The solid lines and filled symbols denote the results for methane, while the dashed lines and open symbols correspond to the results for carbon dioxide. The lines serve as a guide to the eye only.

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